

MICROCRYSTALLINE SILICON: PREPARATION AND PROPERTIES

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ABSTRACT

The plasma parameters controlling the deposition of crystalline and amorphous silicon are discussed and illustrated by examples [1]. The control of the structural properties achieved enables one to perform a systematic study of the transition from the crystalline to the amorphous phase during deposition, as well as a better understanding of some properties of a-Si such as its Raman scattering spectrum [2]. It is shown that the microcrystalline phase becomes thermodynamically unstable with respect to the amorphous phase at a lower limit of the crystallite size of $\sim 30\text{\AA}$ [3]. The effect of reversible adsorption of air constituents on the electrical conductivity, electron spin density and on changes observed in the X-ray Photoelectron Spectra of $\mu\text{c-Si}$ will be described and discussed [4].

The results can be summarized as follows:

A post-deposition exposure of $\mu\text{c-Si}$ films to air results in an irreversible incorporation of oxygen with the formation of Si-O bonds as detected by infrared spectroscopy. In addition, a reversible chemisorption of oxygen (and possibly also of moisture) occurs, resulting in a decrease of the electrical conductivity $\sigma_{\text{R.T.}}$ at room temperature and in an increase of its activation energy. After annealing at $\geq 200^\circ\text{C}$ under high vacuum ($\leq 10^{-6}$ torr), the conductivity increases and its activation energy decreases returning to their original values. The extent of such reversible changes in $\sigma_{\text{R.T.}}$ depends on the deposition conditions, reaching a factor $\geq 10^7$ for samples deposited at low temperatures and only a factor 2-5 for those prepared at $\sim 400^\circ\text{C}$. Table 1 shows, as an example, the effect of the residual pressure of oxygen on the activation energy, E_{act} , on the pre-exponential factor, σ_0 , and on the room temperature conductivity, $\sigma_{\text{R.T.}}$ of a sample deposited at 265°C . The kinetics of both, the

P_{O_2} (torr)	E_{act} (eV)	σ_0 ($\Omega^{-1}\text{cm}^{-1}$)	σ_{RT} ($\Omega^{-1}\text{cm}^{-1}$)
$<10^{-6}$	0.15	2	6×10^3
1	0.37 ₅	58	3×10^5
100	0.89	4×10^8	4.5×10^7

Table 1 (see text)

adsorption and desorption have been studied and will be reported.

Samples exposed to air for a long period of time show an esr signal with $g \pm 2.0055$ and a density of spins of the order of 10^{17}cm^{-3} . This signal vanishes (detection limit $5 \times 10^{15} \text{cm}^{-3}$) after the vacuum annealing as described above; however, it does recover again after long term exposure to air.

An XPS study has revealed only an insignificant shift of the Si 2p binding energy due to the repeated adsorption - desorption process indicating that the observed changes in E_{act} of the electrical conductivity are not dominated by a shift of the Fermi level. The accompanying, reversible appearance of a weak Si 2p signal shifted by $\sim 1.5 \text{eV}$ to lower energy from the pure Si-signal is attributed to a reversible, chemisorbed state on the grain boundaries which obviously influences the charge transport properties of $\mu\text{c-Si}$ as observed by the changes in σ . Further details can be found in the quoted references.

REFERENCES

- (1) S. Vepřek, Z. Iqbal, H.R. Oswald, F.-A. Sarott, J.J. Wagner, J.Physique (1981) (in press)
- (2) Z. Iqbal and S. Vepřek, J.Phys.C: Solid State Phys. (1981) (submitted)
- (3) S. Vepřek, Z. Iqbal, F.-A. Sarott, Phil.Mag. (1981) (submitted)
- (4) S. Vepřek, Z. Iqbal, R.O. Kühne and J. Gimzewski, J.Phys.C: Solid State Phys. (1981) (submitted)