

Ammonia was selected as the nitrogen source. The quenching gases were Ar (5 L/min) and NH₃ (8 L/min).

Metallic Ti powder (40 μm, purity 99.9%, High Purity Chemicals) and Nb powder (20 μm, purity 99%, High Purity Chemicals) were fed into the torch through the powder feeder by Ar carrier gas (3 L/min). The powder feed rate was fixed at 300 mg/min.

Table 1. Experimental operating conditions for the preparation of Ti_{1-x}Nb_xN nanoparticles.

Input power [kW]	20
RF frequency [MHz]	4
Pressure [kPa]	101.3
Sheath gas rate [L/min]	60 (Ar)
Inner gas rate [L/min]	5 (Ar)
Carrier gas rate [L/min]	3 (Ar)
Feed rate [mg/min]	300
Quenching gas rate [L/min]	5 (Ar)
	8 (NH ₃)
Injection position	15 cm

The crystal structure and phase identification of the synthesized nanoparticles were determined through powders X-ray diffraction (XRD, Rigaku Multiflex), operating with a Cu Kα source (λ = 0.1541 nm).

3. Thermodynamic Calculation

Thermodynamic calculations were conducted with software *FactSage 8.1*. The calculations were performed for temperatures up to 5,000 K and pressure of 101.3 MPa for the compositions of the technological mixtures. The Gibbs free energy diagram for the decomposition of N₂ and NH₃ are shown in Fig. 2. N₂ molecules dissociate into N radicals above approximately 7,000 K. NH₃ dissociates into NH radicals and NH₂ radicals above approximately 3,000 K. As shown in Figs. 3 and 4, the negative Gibbs free energy of Ti/Nb and NH₂/NH radicals below 5,000 K

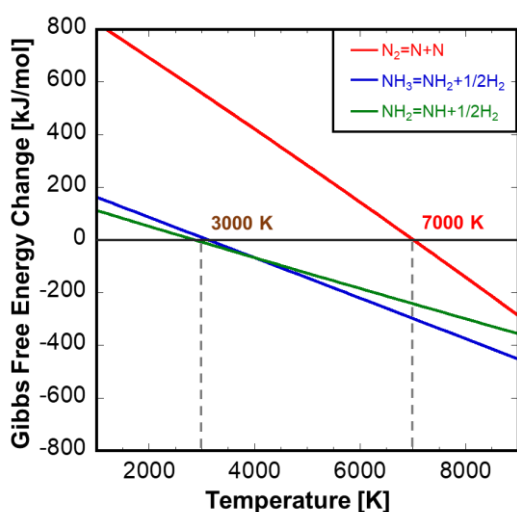


Fig. 2 Gibbs free energy diagram for the decomposition of N₂ and NH₃.

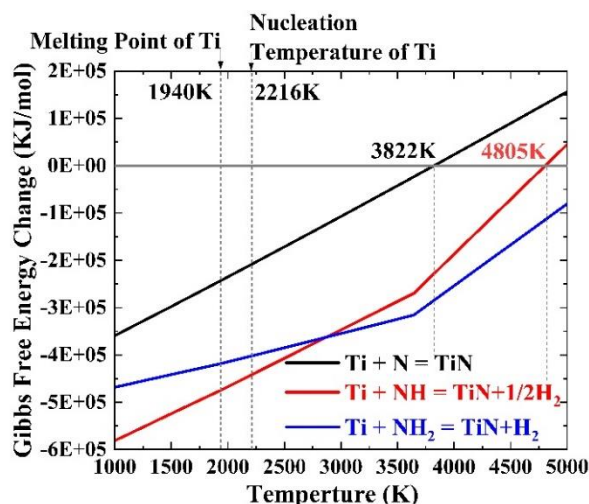


Fig. 3 Relationship between Gibbs free energy change and nucleation point in Ti-N system.

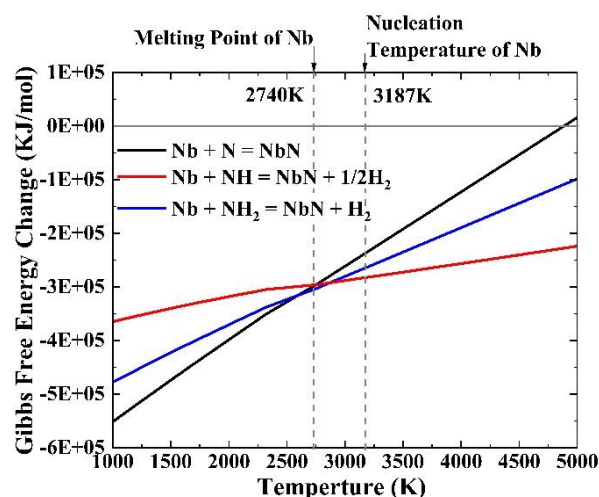


Fig. 4 Relationship between Gibbs free energy change and nucleation point in Nb-N system.

indicates high reactivity. Ammonia can provide a large amount of NH₂ and NH radicals at 3000 ~ 5000 K. These radicals can make Ti/Nb sufficiently nitrided. Therefore, NH₃ was selected as nitrogen source.

4. Results and discussion

The XRD patterns of Ti_{1-x}Nb_xN nanoparticles with different molar ratios of Ti/Nb are shown in Fig. 5. Titanium nitride was successfully synthesized without Nb injection. Main peaks were assigned to be a cubic TiN structure with a space group of *Fm-3m*. The diffraction peaks shift to the left with Nb injection and it is more obvious at x=0.75. The shift of diffraction peaks suggests the injection of Nb into TiN because the ionic radius of Nb (0.72 Å) is larger than the ionic radius of Ti (0.67 Å). The diffraction peaks are lowered after adding Nb to TiN. This implies that Nb addition plays a significant role in the structural properties of TiN. The results also indicated a small diffraction peak of NbN_x with Nb injection. The content of NbN_x increased with the increase of Nb content.

The obtained results clearly revealed that $\text{Ti}_{1-x}\text{Nb}_x\text{N}$ nanoparticles with high purity are successfully prepared by RF induction thermal plasma.

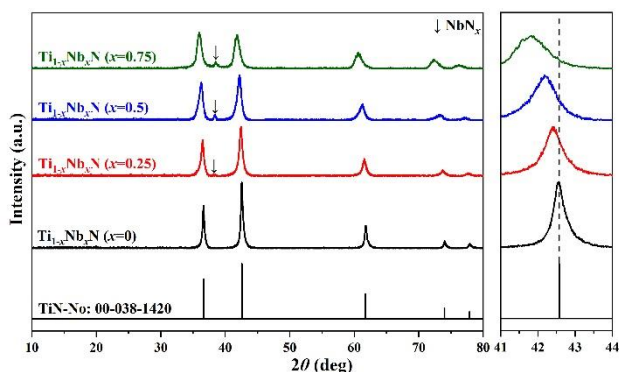


Fig. 5 XRD patterns of $\text{Ti}_{1-x}\text{Nb}_x\text{N}$ nanoparticles with different molar ratios of Ti/Nb

5. Conclusion

Titanium niobium nitrides nanoparticles with high purity were successfully prepared by RF induction thermal plasma. Metallic Ti and Nb powders were used as raw materials. Main peaks were assigned to be a cubic TiN structure with a space group of $Fm-3m$. Ammonia injection into plasma tail as counter flow is a suitable method for transition metal nitridation. Induction thermal plasma can provide a highly efficient way to synthesize ternary transition metal nitrides to apply for large-scale industrial production.

6. Acknowledgements

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7. References

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